Nuclear Materials from Experiments and Atomistic Simulations

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Majority of HLW (High Level Nuclear Waste) consists of directly stored UO₂-based spent nuclear fuel or its vitrified form. In order to provide a solid scientific basis of nuclear waste management and disposal we try to understand the molecular-scale originating properties of various relevant nuclear materials, including waste and waste forms, and their long-term performance. In this contribution, we report the results of joint atomistic modeling and experimental studies of HLW related materials such as nuclear glass and Cr-doped UO_2 -based model systems.

Atomistic simulations

In our work we apply a variety of atomistic modeling methods. The simulations of glasses are performed with the method of classical molecular dynamics. Here we apply force fields^[1] to describe the interactions between atoms. We use LAMMPS and GULP codes^[2] for that purpose. The simulations of solid phases are based on *ab initio* Density Functional Theory (DFT). For *f*-elements we use the DFT + U method but with the Hubbard U parameter values derived from first principles using the linear response method^[3-4].

Simulations of mixed uranium-oxides





Atomistic simulations and thermodynamic modeling

Simulations of borosilicate glasses

Measurements of Young's modulus and Hardness by nano-indentation^[5-6]

Tab. 1 The compositions of BS1-8 Kr-NBS1 Fitting line Mol% BS3 BS4 BS5 BS6 BS7 BS8 48.0 52.8 58.7 62.9 66.0 68.4 SiO₂ 44.0 40.0 35.2 29.3 25.1 22.0 19.6 B_2O_3 44.0 12.00 Na₂O Xe-NBS1 - - Fitting line • BS1-8 BS1-8 Pristine

UO₂-based materials could undergo structural and chemical change. In new

phases U(IV) could oxidise from U(IV) to U(VI), producing the risk of fast dissolution. Here by combination of HR-XANES and atomistic simulations we investigate electronic structure of mixed uranium-oxides^[8]. The best match to the experiment is obtained with DFT+U method, the derived Hubbard U parameter and Wannier representation of *f* orbitals.

Simulations of Cr-doped UO

Addition of Cr to UO_2 enlarges the grain sizes and improves the performance of the nuclear fuel(e.g. by limiting fission gas release). We want to understand how Cr is incorporated into the bulk-UO₂ and in which oxidation state. Is it Cr(III) as claimed by experimental studies^[9]?

 Tab. 2 Relative lattice parameter



△ Models from Farges's work [12



Results:

Increase of Young's modulus and Hardness with Si content.

Measurements of change of Young's modulus and Hardness under irradiation.

Clear decrease in Hardness, amorphization at ~0.1 dpa. Experimental data from [5-6]

Open question: What causes the observed change?



Test case: Simulations correctly predict stored energy and volume change of





- DFT+U simulations show that Cr incorporates as Cr(II) and the charge is balanced by O vacancies or formation of U(V).
- The thermodynamically favorable structure composes of associated pair of Cr(II) atom and O vacancy (neighboring site).
- XANES is used to judge the oxidation states of elements (*e.g.* Cr in UO₂).
- The re-evaluation of the experimental XANES data shows presence of Cr(II). This structure fits the experimental data: Cr solubility, XANES and change in lattice parameter.^[13]

Simulations of MUO_{4-x} (Secondary phases)

Good description of defects and their formation

We investigate T-driven O-defects formation & phase transformation in MUO_{4-x} phases. Our simulations reproduce measured defect formation energies^[14] well. δ-CaUO_{4-y} \square α -SrUO. $\blacktriangle \alpha$ -Sr_{0.4}Ca_{0.6}UO₄-

Simulations correctly predict the critical amorphization dose at ~ 0.1 dpa and Young's modulus of irradiated NBS glass. Reason for discrepancy observed for pristine glass is currently being investigated.





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 $3.0 \hat{a}$

Calculations

Veutron scattering data

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